

Letter

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# TEM and ED confirmation of conversion of 3D and 2D perovskite-type into fluorite-type structure

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#### ABSTRACT

In this paper, we confirm the transformation of three-dimensional (3D) In-doped perovskite-like  $BaCe_{1-x}In_xO_{3-\delta}$  (x=0.1; 0.2) and layered (two-dimensional) perovskite-related  $Sr_2Ce_{1-x}Y_xO_{4-\delta}$  (x=0.1; 0.2) into corresponding fluorite-type  $Ce_{1-x}M_xO_{2-\delta}$  (M=In, Y) under  $CO_2$  mediated reaction at elevated temperature, respectively, using high-resolution transmission electron microscopy (HRTEM) and electron diffraction (ED) studies. HRTEM images confirmed expected nano-sized materials and ED study showed the formation of a single-phase fluorite-type CeO<sub>2</sub> structure and absence of ED patterns resulting from the perovskite-like precursors and potential reaction by-products such as  $In_2O_3$  and  $Y_2O_3$  after the  $CO_2$  reaction, which is consistent with ex situ and in situ powder X-ray diffraction (PXRD), and strongly support the proposed structural transformation reactions.

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#### 1. Introduction

Design and preparation of solid-state functional property materials for specific applications, including fuel cells, batteries, sensors, thermoelectrics, photovoltaics, dielectrics, electronics, and catalysts, have been routine work for several material chemists and scientists [1–6]. Solid-state materials are being commonly synthesized using a conventional mixed-oxide route, well known as a ceramic method, at elevated temperatures. Several low-temperature methods, also popularly called as chimiedouce synthesis, involve preparation of precursor materials by ceramic method and subsequently transformed into desired and designed novel metastable functional materials [3-9]. A wellknown example is the synthesis of new modification of TiO<sub>2</sub> by a mild-dehydration of  $H_2Ti_4O_9$ · $H_2O$ . The precursor  $H_2Ti_4O_9$ · $H_2O$  was prepared from the K-analogue K<sub>2</sub>Ti<sub>4</sub>O<sub>9</sub> (synthesized by ceramic reaction) by ion-exchange reaction [7]. Other significant examples are ion-exchange and double ion-exchange (metathesis reaction) of a large family of layered (2D) perovskite-like Dion-Jacobson (DJ) and Ruddlesden-Popper (RP)-type structures and inter-conversion of RP phase into Aurivillius structure and vice-versa [8-11]. Numerous metastable allotropes compounds were also prepared by gentle dehydration of proton and/or hydrated metal oxides in air or vacuum. Another synthesis method utilizes  $H_2$  or inert atmospheres to prepare compounds containing lower valence cations than that of prepared in ambient condition.

Our group has employed a new two steps synthesis to stabilize indium-doped CeO<sub>2</sub> structure, which was not attainable by a regular mixed-oxide solid-state reaction at 800–1500 °C [12–15]. This approach involves leaching of alkaline from the perovskite and its related structure precursors under CO<sub>2</sub> at elevated temperatures and subsequent acid washing. In the literature, the chemical stability of doped perovskite-type ABO<sub>3</sub> (A=Sr, Ba, B=Ce, Zr) under CO<sub>2</sub> at elevated temperatures was well documented [16–24]. For example, Y<sub>2</sub>O<sub>3</sub>-doped BaCeO<sub>3</sub> (BCY) was found to be chemically unstable in pure CO<sub>2</sub> at elevated temperature according to the reaction BaCe<sub>1-x</sub>Y<sub>x</sub>O<sub>2-\delta</sub>(s) + CO<sub>2</sub>(g)  $\xrightarrow{\Delta}$  BaCO<sub>3</sub>(s) + Ce<sub>1-x</sub>Y<sub>x</sub>O<sub>3-\delta</sub>(s) and partial substitution of Ti and or Zr for Ce in BCY improved the chemical stability in CO<sub>2</sub> [12,13,17,25–30]. Recently, Xie and others have investigated the effect of Sn, Ta and Nb substitution for Ce in BCY on chemical stability and proton conductivity [25–30].

It is very important to note that acceptor-doped ceria based materials are being considered as the potential electrolyte materials for the intermediate temperature solid oxide fuel cells (IT-SOFCs). Particularly,  $Ce_{1-x}M_xO_{2-\delta}$  (M = Y, Sm, Gd) have drawn much attention because of their high oxide ion conductiv-

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ity compared to  $Y_2O_3$ -doped  $ZrO_2$  and SrO- and MgO-doped perovskite-like LaGaO<sub>3</sub> (LSGM) [31]. Furthermore, the doped CeO<sub>2</sub>-based electrolytes appear to be chemically stable against reaction with Ni anode as well as several perovskite-type structure cathode materials such as  $La_{1-x}Sr_xMO_3$  (M = Mn, Fe, Co) [32]. Accordingly, understanding of structure and morphology (particle size) of the fluorite-type  $Ce_{1-x}M_xO_{2-\delta}$  (M = In, Y) compounds prepared from perovskites under CO<sub>2</sub> atmosphere is very useful for further development of ceria electrolytes. Also, the present work provides answer to a key question about phase purity of the fluorite-type product obtained after CO<sub>2</sub> reaction using electron diffraction (ED). In this letter, we confirm the formation a single-phase fluorite-type solid solution  $Ce_{1-x}M_xO_{2-\delta}$  using high-resolution transmission electron microscopy (HRTEM) combined with ED and PXRD.

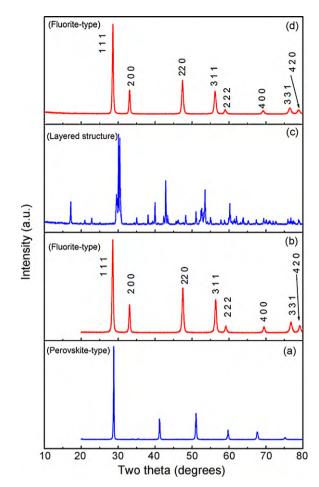
#### 2. Experimental details

Ce<sub>1-x</sub>M<sub>x</sub>O<sub>2-δ</sub> (M=In, Y; x=0.1; 0.2) was synthesized by CO<sub>2</sub> mediated process from the perovskite-type BaCe<sub>1-x</sub>In<sub>x</sub>O<sub>3-δ</sub> and layered Sr<sub>2</sub>Ce<sub>1-x</sub>Y<sub>x</sub>O<sub>4-δ</sub>, respectively, at elevated temperature of 800 °C and subsequent acid washing using dilute mineral acids [11–13]. The precursors BaCe<sub>1-x</sub>In<sub>x</sub>O<sub>3-δ</sub> and Sr<sub>2</sub>Ce<sub>1-x</sub>Y<sub>x</sub>O<sub>4-δ</sub> were prepared by a conventional ceramic method in air with final sintering temperature at 1500 °C and 1350 °C, respectively for 24 h, using appropriate amounts of high purity SrCO<sub>3</sub> (99+%, VWR International), BaCO<sub>3</sub> (99+%, VWR International), CeO<sub>2</sub> (99.5%, Alfa Aesar), and In<sub>2</sub>O<sub>3</sub> (99.9%, Alfa Aesar). As-prepared samples after CO<sub>2</sub> reaction and subsequent acid washed were characterized using a powder X-ray diffraction (PXRD) (Bruker D8 powder X-ray diffractometer; CuKα, 40 kV, 40 mA) at room temperature.

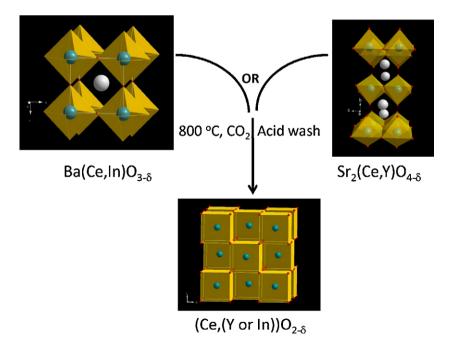
Further characterization involves use of transmission electron microscopy (TEM) coupled with electron diffraction (ED). It was done on a FEI Tecnai F20 FEG-TEM (FEI, Eindhoven, The Netherlands) equipped with a Gatan Imaging Filter and a Gatan 860 GIF 2001 CCD of  $1024 \times 1024$  resolution. Standard TEM was done at magnifications between 8000 and 67,000. HRTEM was performed at magnifications between 940,000 and 1,350,000. The diffraction patterns were taken at a camera length of 400 mm and directly after taking each individual diffraction pattern of the sample, it was exchanged by a reference sample of amorphous gold without changing any settings on the microscope. The diffraction rings of this amorphous gold were used to confirm a proper calibration in diffraction mode of the equipment.

#### 3. Results and discussion

PXRD (Fig. 1) confirmed the formation of fluorite-type structure for  $Ce_{1-x}M_xO_{2-\delta}$  (M = In, Y; x = 0.1; 0.2) from the corresponding  $BaCe_{1-x}M_xO_{3-\delta}$  under  $CO_2$  at elevated temperatures, which was



**Fig. 1.** PXRD patterns of (a) as-prepared perovskite-like structure  $BaCe_{0.9}In_{0.1}O_{2.95}$ , (b) after heating the sample "a" at 800 °C in CO<sub>2</sub> for 12 h and subsequently treated it with dilute acid and dried in ambient air, (c) K<sub>2</sub>NiF<sub>4</sub>-related layered perovskite-type structure of as-prepared Sr<sub>2</sub>Ce<sub>0.9</sub>Y<sub>0.1</sub>O<sub>3.95</sub>, (d) after heating the sample "c" at 800 °C in CO<sub>2</sub> for 12 h and subsequently treated it with dilute acid and dried in ambient air.



**Fig. 2.** Idealized crystal structure showing the formation of In and or Y-doped CeO<sub>2</sub> from the perovskite-like structure doped BaCeO<sub>3</sub> and perovskite-related layered structure doped Sr<sub>2</sub>CeO<sub>4</sub> in CO<sub>2</sub> at elevated temperature and subsequent acid washing [14,15].

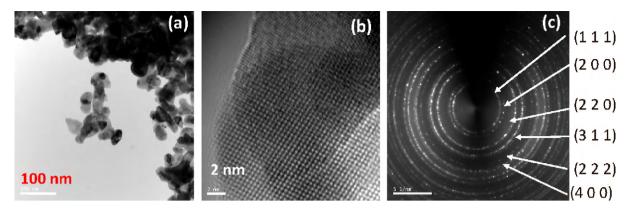


Fig. 3. TEM and HRTEM images of as-prepared (CO<sub>2</sub> capture method) powdered Ce<sub>0.9</sub>In<sub>0.1</sub>O<sub>1.95</sub> from the corresponding 10 mol% In-doped BaCeO<sub>3</sub>: (a) 100 nm scale, (b) 2 nm scale and (c) selected area electron diffraction pattern.

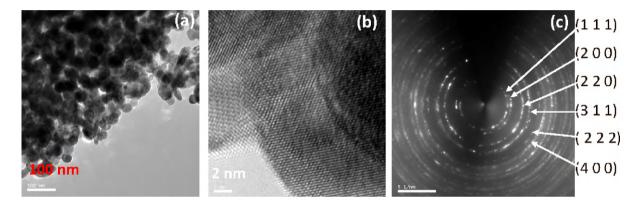


Fig. 4. TEM and HRTEM images of as-prepared (CO<sub>2</sub> capture method) powdered Ce<sub>0.8</sub>In<sub>0.2</sub>O<sub>1.9</sub> from the corresponding 20 mol% In-doped BaCeO<sub>3</sub>: (a) 100 nm scale, (b) 2 nm scale and (c) selected area electron diffraction pattern.

found to be consistent with ex situ and in situ X-ray studies [12,14,15]. The PXRD of CO<sub>2</sub> mediated synthesized Ce<sub>0.9</sub>In<sub>0.1</sub>O<sub>1.95</sub>, Ce<sub>0.8</sub>In<sub>0.2</sub>O<sub>1.9</sub>, Ce<sub>0.9</sub>Y<sub>0.1</sub>O<sub>1.95</sub> and Ce<sub>0.8</sub>Y<sub>0.2</sub>O<sub>1.9</sub> could be indexed on a cubic fluorite-type lattice constant of 5.398(1)Å, 5.393(1)Å and 5.421(2)Å, respectively. Fig. 2 shows the idealized crystal structure of the conversion of a perovskite and perovskite-related structure materials into corresponding metal-doped fluorite-type structure. We have also performed energy dispersive X-ray spectroscopy (EDX) for the synthesized powder materials on the large surface area to confirm that the investigated samples were indeed the doped CeO<sub>2</sub> materials. The EDX data were found support the proposed chemical composition and is consistent with our earlier work [14,33].

In order to validate our proposed structural transformation reactions, we have performed HRTEM and ED studies on four different fluorite-type compounds such as  $Ce_{0.9}In_{0.1}O_{1.95}$ ,  $Ce_{0.8}In_{0.2}O_{1.9}$ ,  $Ce_{0.9}Y_{0.1}O_{1.95}$ , and  $Ce_{0.8}Y_{0.2}O_{1.9}$  prepared by CO<sub>2</sub> mediated reactions. In Figs. 3 and 4, we show typical HRTEM images of 10 mol% and 20 mol%  $In_2O_3$ -doped CeO<sub>2</sub> synthesized from the corresponding In-doped BaCeO<sub>3</sub> at elevated temperature under CO<sub>2</sub>, respectively. Shown in Figs. 5 and 6 are the corresponding data for the 10 mol% and 20 mol%  $Y_2O_3$ -doped CeO<sub>2</sub> synthesized from the layered structure  $Sr_2CeO_4$ .

As anticipated, the selected area ED patterns showed the lines only due to fluorite-type structure. We have successfully assigned all the diffraction lines into  $CeO_2$  structure and are

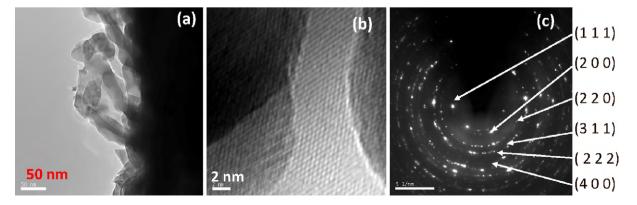
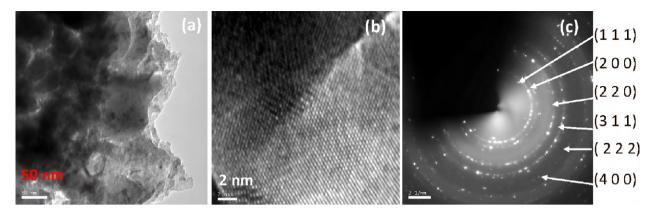


Fig. 5. TEM and HRTEM images of as-prepared (CO<sub>2</sub> capture method) powdered Ce<sub>0.9</sub>Y<sub>0.1</sub>O<sub>1.95</sub> from the corresponding 10 mol% Y-doped Sr<sub>2</sub>CeO<sub>4</sub>: (a) 100 nm scale, (b) 2 nm scale and (c) selected area electron diffraction pattern.



**Fig. 6.** TEM and HRTEM images of as-prepared (CO<sub>2</sub> capture method) powdered Ce<sub>0.8</sub>Y<sub>0.2</sub>O<sub>1.9</sub> from the corresponding 20 mol% Y-doped Sr<sub>2</sub>CeO<sub>4</sub>: (a) 100 nm scale, (b) 2 nm scale and (c) selected area electron diffraction pattern.

consistent with powder XRD data. We did not observe any main diffraction lines due to cubic In<sub>2</sub>O<sub>3</sub> (e.g., 222 plane;  $d \approx 2.92$  Å) joint committee on powder diffraction standard (JCPDS Card 060-416; JCPDS card no. 65-3170), hexagonal In<sub>2</sub>O<sub>3</sub> (JCPDS card no. 22-0336), and Y2O3 (JCPDS card no. 25-1200) as well as the starting materials in the product, confirming the proposed transformation reaction. HRTEM shows the formation of nano-sized material which was found to consistent with several authors report of nano-sized CeO<sub>2</sub> [34-36]. The Y-sample was found have slightly larger particle size than that of the In-doped samples. Mechanistically, at elevated temperatures, the perovskite-like  $BaCe_{1-x}In_xO_{3-\delta}$  (x=0.1; 0.2) and layered perovskite-related  $Sr_2Ce_{1-x}Y_xO_{4-\delta}$  (x = 0.1; 0.2) structures transform into their corresponding fluorite-type  $Ce_{1-x}In_xO_{2-\delta}$  and  $Ce_{1-x}Y_xO_{2-\delta}$  and BaCO<sub>3</sub>. These fluorite-type phases of In-doped CeO<sub>2</sub> are stable up to  $\sim$ 800 °C [12,15].

#### 4. Conclusions

In summary, the ED and TEM investigations strongly support the formation of In and Y-doped fluorite-type  $Ce_{1-x}In_xO_{2-\delta}$  and  $Ce_{1-x}Y_xO_{2-\delta}$  from three-dimensional (3D) perovskite-like  $BaCe_{1-x}In_xO_{3-\delta}$  (x=0.1; 0.2) and layered (two-dimensional) perovskite-related  $Sr_2Ce_{1-x}Y_xO_{4-\delta}$  (x=0.1; 0.2), respectively, under CO<sub>2</sub> mediated reaction at elevated temperature. The absence of any ED pattern that would correspond to the perovskite-like precursors and potential reaction by-products such as  $In_2O_3$  and  $Y_2O_3$  after the CO<sub>2</sub> reaction, evidently confirms the formation of a single-phase fluorite-type structure, which is consistent with ex situ and in situ powder X-ray diffraction (PXRD) analysis. The HRTEM images also confirmed expected nano-sized materials.

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#### References

- [1] M.G. Kanatzidis, K.R. Poeppelmeier, Prog. Solid State Chem. 36 (2008) 1.
- [2] A. Stein, S.W. Keller, T.E. Mallouk, Science 259 (1993) 1558.
- [3] C.N.R. Rao, J. Gopalakrishnan, New Directions in Solid State Chemistry, 2nd ed., Cambridge University Press, Cambridge, 1997.
- [4] S.E. Dann, Reactions and Characterization of Solids, Wiley InterScience, New York, 2002.
- [5] C.N.R. Rao, Chemical Approaches to the Synthesis of Inorganic Materials, John Wiley and Sons, New York, 1994.
- [6] C.B. Alcock, J. Alloys Compd. 197 (1993) 217.
- [7] R. Marchand, L. Brohan, M. Tournoux, Mater. Res. Bull. 15 (1980) 1129.
- [8] R.E. Schaak, T.E. Mallouk, Chem. Mater. 14 (2002) 1455.
- [9] J. Gopalakrishnan, Chem. Mater. 7 (1995) 1265.
- [10] W. Sugimoto, M. Shirata, Y. Sugahara, K. Kuroda, J. Am. Chem. Soc. 121 (1999) 11601.
- [11] Y. Tsunoda, M. Shirata, W. Sugimoto, Z. Liu, O. Terasaki, K. Kuroda, Y. Sugahara, Inorg. Chem. 40 (2001) 5768.
- [12] F. Trobec, V. Thangadurai, Inorg. Chem. 47 (2008) 8972.
- [13] B.R. Sneha, V. Thangadurai, J. Solid State Chem. 180 (2007) 2661.
- [14] R.G. Gerlach, S.S. Bhella, V. Thangadurai, Inorg. Chem. 48 (2009) 257.
- [15] S.S. Bhella, S. Shafi, F. Trobec, M. Bieringer, V. Thangadurai, Inorg. Chem. 49 (2010) 1699.
- [16] K.D. Kreuer, Solid State Ionics 97 (1997) 1.
- [17] K.D. Kreuer, Annu. Rev. Mater. Res. 33 (2003) 333.
- [18] H. Iwahara, Y. Asakura, K. Katahira, M. Tanaka, Solid State Ionics 168 (2004) 299.
- [19] C.W. Tanner, A.V. Virkar, J. Electrochem. Soc. 143 (1996) 1386.
- [20] S.V. Bhide, A.V. Virkar, J. Electrochem. Soc. 146 (1999) 2038.
- [21] S.V. Bhide, A.V. Virkar, J. Electrochem. Soc. 146 (1999) 4386.
- [22] H. Matsumoto, Y. Kawasaki, N. Ito, M. Enoki, T. Ishihara, Electrochem. Solid-State Lett. 10 (2007) B77.
- [23] A.K. Azad, J.T.S. Irvine, Solid State Ionics 178 (2007) 635.
- [24] G. Chiodellia, L. Malavasi, C. Tealdi, S. Barison, M. Battagliarin, L. Doubova, M. Fabrizio, C. Mortalo, R. Gerbasi, J. Alloys Compd. 470 (2009) 477.
- [25] K. Xie, R. Yan, X. Liu, J. Alloys Compd. 479 (2009) L40.
- [26] R. Yan, Q. Wang, G. Chen, W. Huang, K. Xie, Ionics 15 (2009) 749.
- [27] K. Xie, R. Yan, X. Liu, J. Alloys Compd. 479 (2009) L36.
- [28] R. Yan, Q. Wang, K. Xie, Ionics 15 (2009) 501.
- [29] R. Yan, G. Chen, F. Wang, Q. Wang, W. Huang, J. Alloys Compd. 486 (2009) L10.
- [30] K. Xie, R. Yan, X. Xu, X. Liu, G. Meng, Mater. Res. Bull. 44 (2009) 1474.
- [31] J.B. Goodenough, Annu. Rev. Mater. Res. 33 (2003) 91.
- [32] S.B. Adler, Chem. Rev. 104 (2004) 4791.
- [33] S.S. Bhella, L.M. Kuti, Q. Li, V. Thangadurai, Dalton Trans. (2009) 9520.
- [34] C. Sun, H. Li, H. Zhang, Z. Wang, L. Chen, Nanotechnology 16 (2005) 1454.
- [35] C. Hu, Z. Zhang, H. Liu, P. Gao, Z.L. Wang, Nanotechnology 17 (2006) 5983.
- [36] A.I.Y. Tok, F.Y.C. Boey, Z. Dong, X.L. Sun, J. Mater. Process. Technol. 190 (2007) 217.